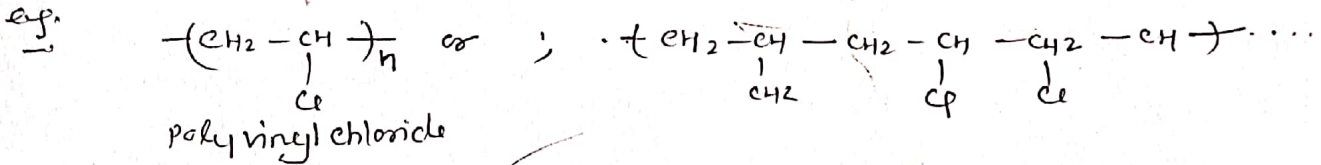
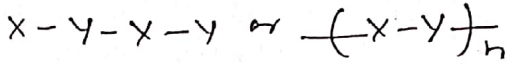
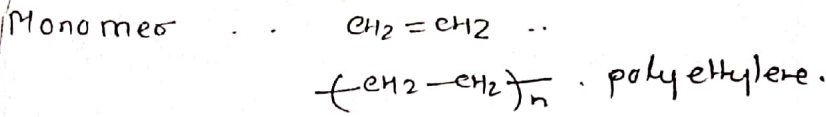
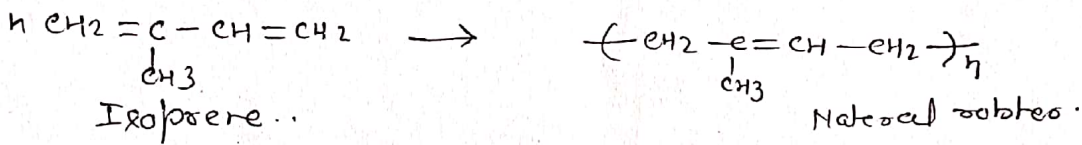


Polymers:- Introduction, classification, structure of organic and inorganic polymers of industrial importance, liquid crystalline and star polymers etc. (1)

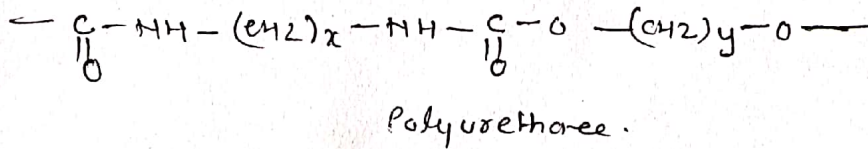
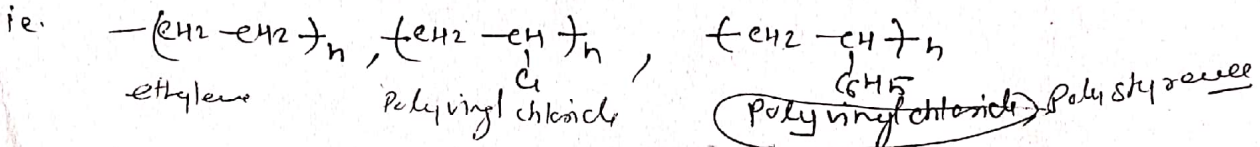


Classification of Polymer

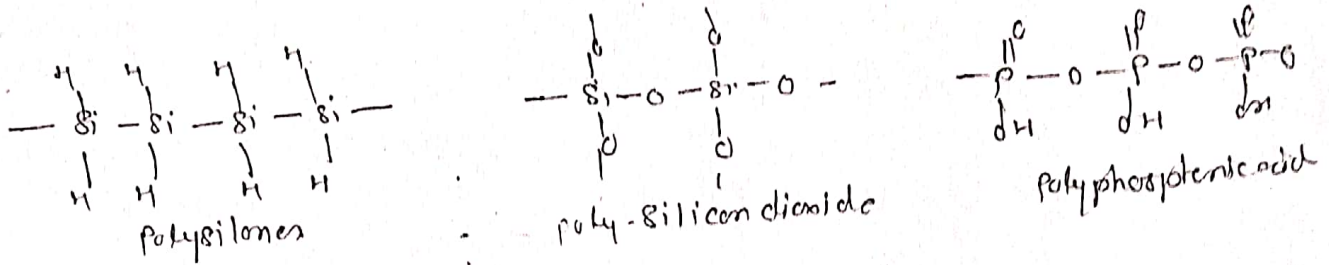
① Natural Polymer  $\rightarrow$  occur in nature i.e. they have vegetable or an animal origin, i.e. rubber, proteins, cellulose & polysaccharides, wood and silk. Cellulose and starch are the polymers of glucose. Proteins are the macromolecules containing a large number of amino acids. Natural rubber contains isoprene (2-methyl-1,3-butadiene) repeat unit.



Synthetic or Organic Polymer:- i.e. man-made polymer. i.e. contain carbon, hydrogen, oxygen, nitrogen, sulphur and halogen atoms in the backbone or main chain.

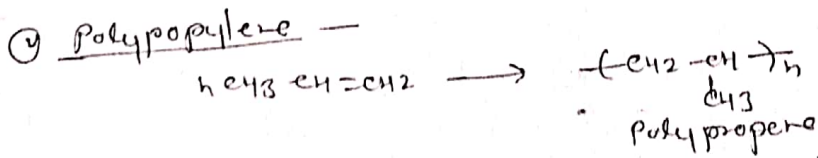
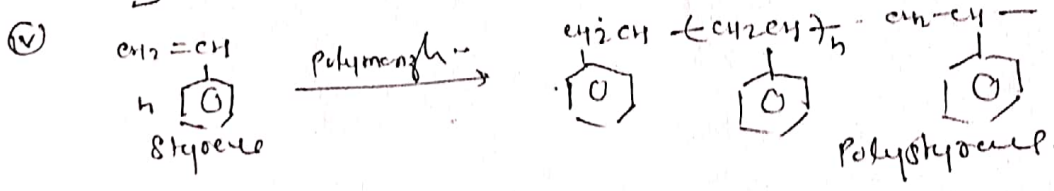
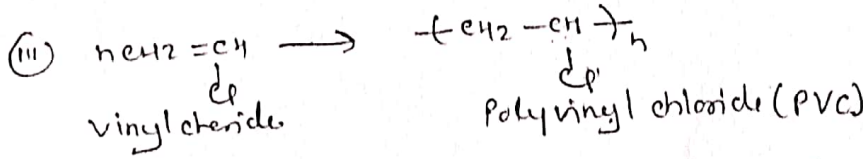
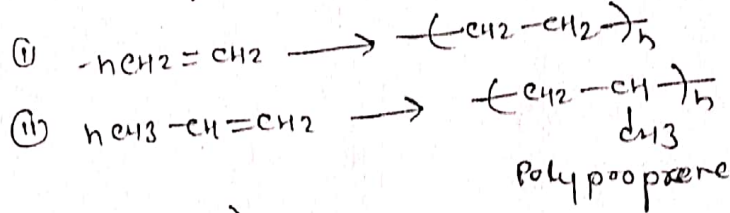


⑩ Inorganic Polymers :- These polymers contain no carbon atoms. i.e. atoms joined by chemical bonds. weak inter-molecular forces act between the chains.



Addition Polymers :-

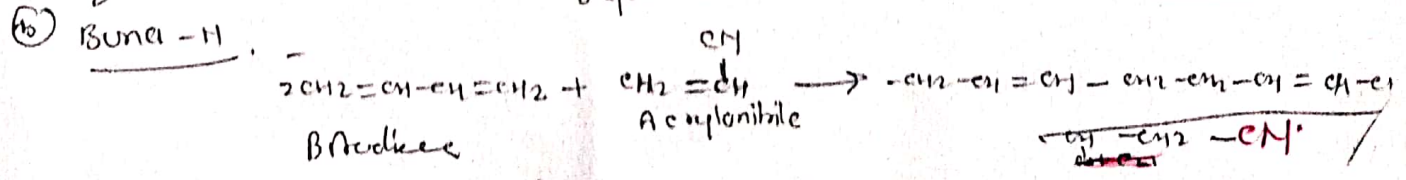
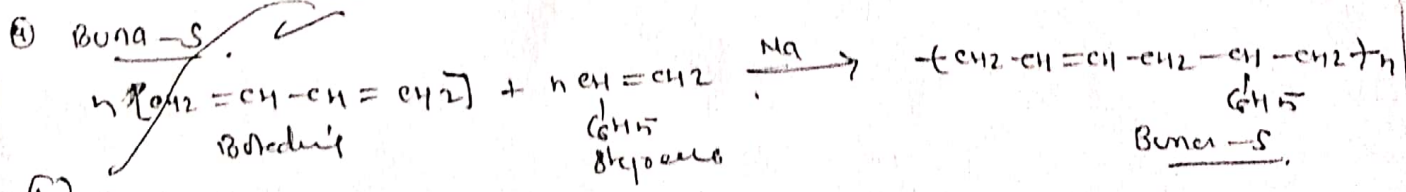
Polymers



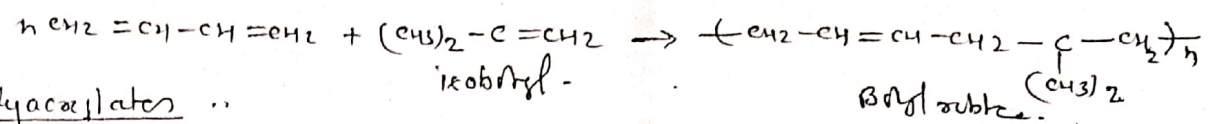
Properties & Uses :- Polypolyene is quite resistant to acid and alkalis. It has high tensile strength, it is harder and stronger than polyethylene.

Polydienes :- These polymers are obtained from the polymerization of unsaturated hydrocarbons with two double bonds.

- ① Synthetic Rubbers :-
- (a) Buna Rubbers
- (b) Butyl Rubbers

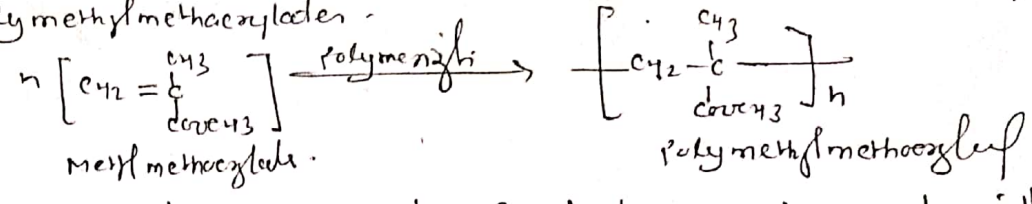


① Bdyl Rubber :

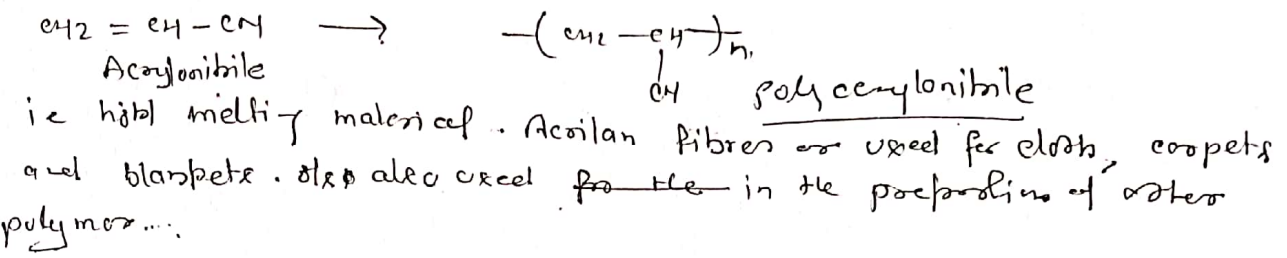
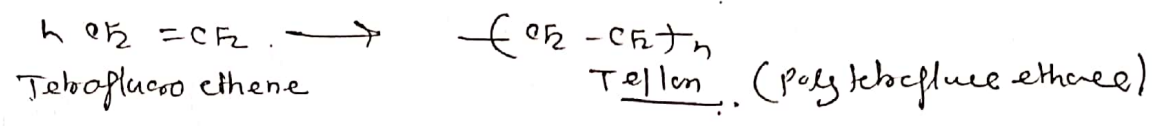


② Polyacrylates ..

③ Polymethylmethacrylates -

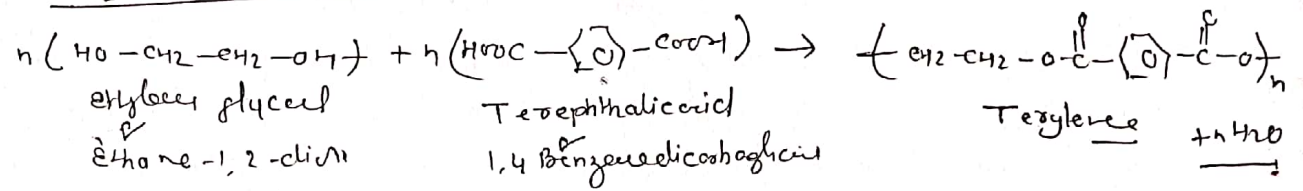


④ is hard, transparent with high optical clarity - it has great resistance to effect of light and ageing. It is useful for lenses, transparent domes and skylights, dentures aircraft windows and protective coatings. Its commercial names are ~~Acryl~~ Lucite, plexiglass, acrylite and perspex.

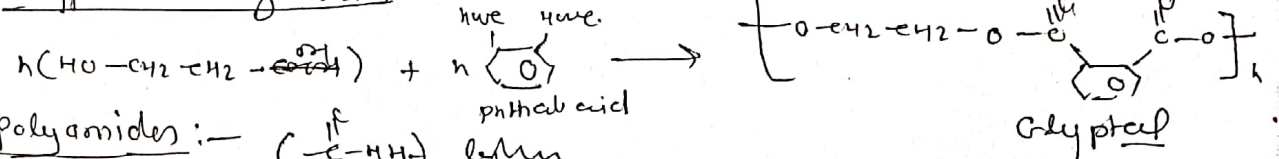


CONDENSATION POLYMERS ① polyester -  $(\text{R}-\overset{\text{O}}{\parallel}{\text{C}}-\text{O}-\text{R}')_n$

② Terylenes (a decron) :-

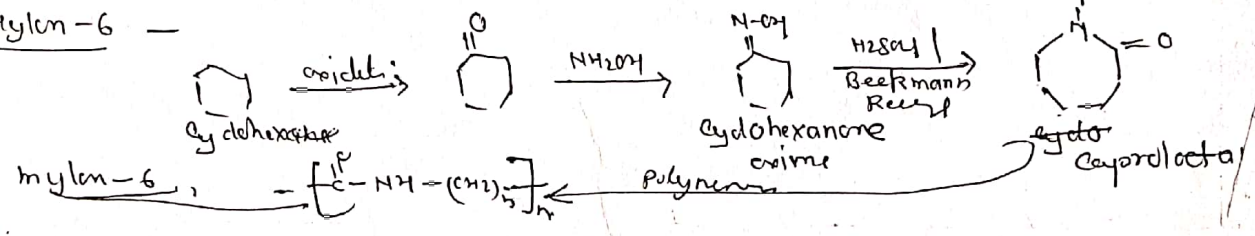


③ Glyptal or alkyl resin -



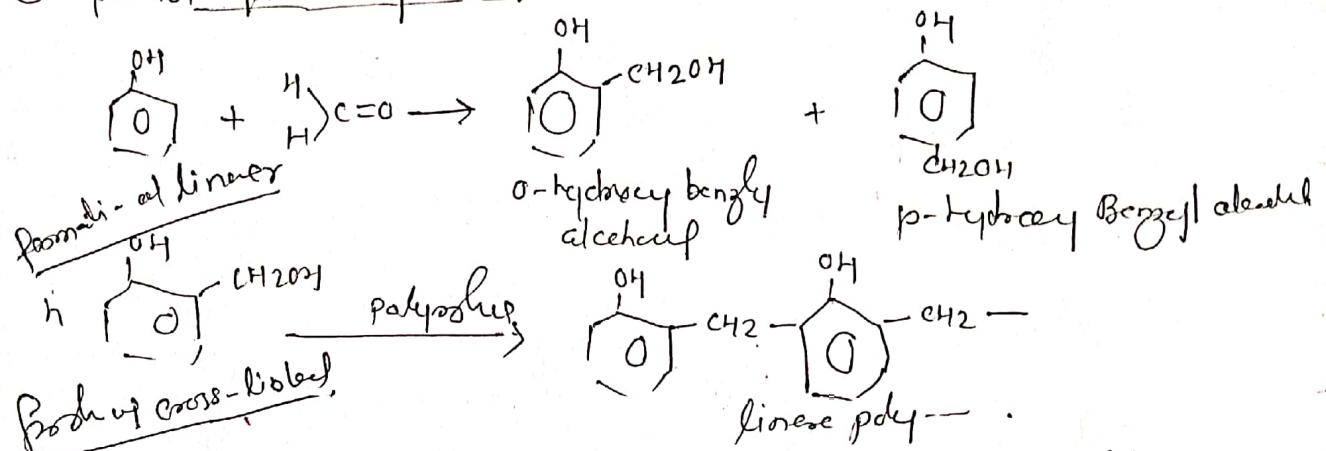
④ Polyamides :-  $\left( \overset{\text{O}}{\parallel}{\text{C}}-\text{NH} \right)_n$

⑤ Nylon-6 -



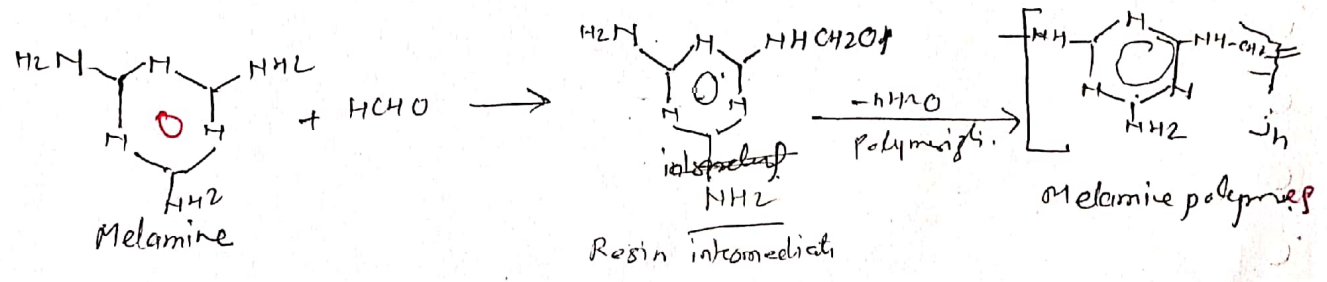
(4) Formaldehyde Resins: -- Formaldehyde resins are typical thermosetting plastics. This class of plastics include phenol-formaldehyde, urea-formaldehyde and melamine-formaldehyde resins. On heating these resins become highly cross-linked thereby forming infusible, and insoluble product.

(a) phenol-formaldehyde resins



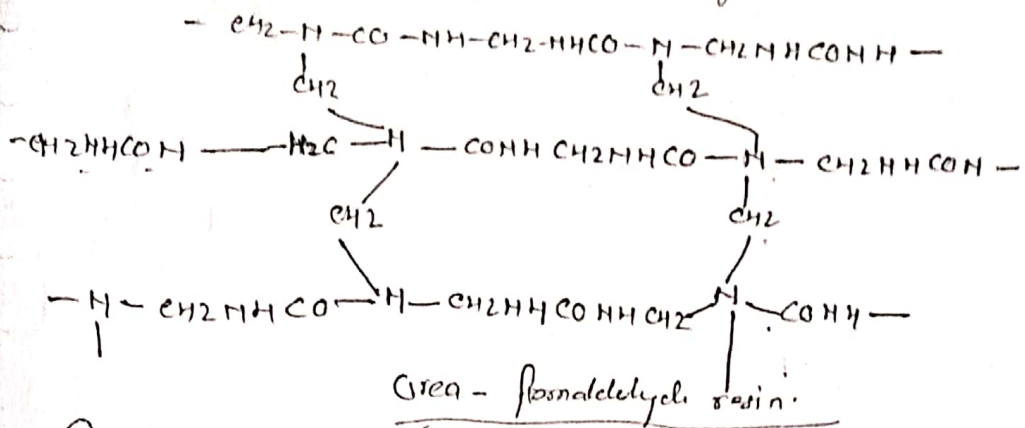
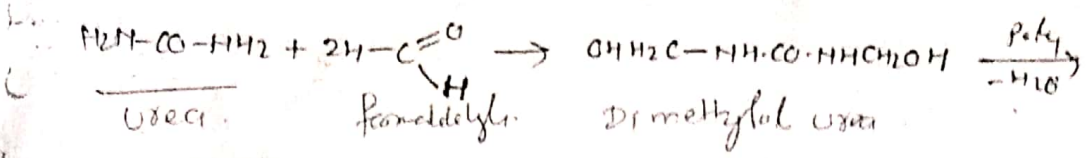
High degree of polymerization leads to the formation of hard bakelite which is used for making combs, fountain-pen, barrels, phonograph records, electric goods, Formica table-top and gear wheels for steel rolling and paper mills, protective casing of electrical apparatus. They can be laminated with paper, leather in order to obtain products with high resistance to impact.

(b) Melamine-formaldehyde resins



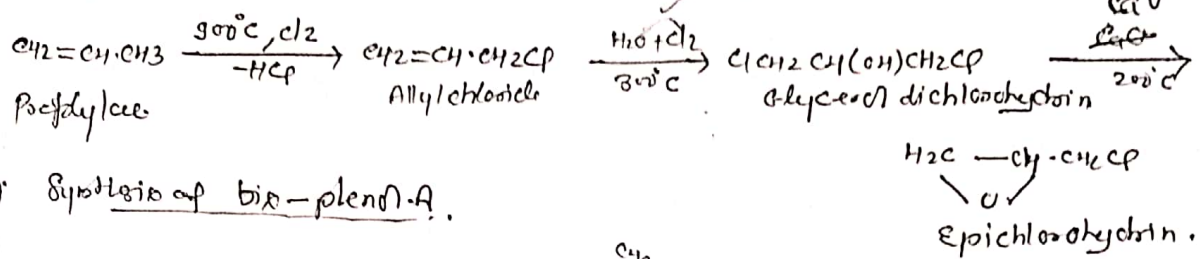


(4) Urea formaldehyde - (Urea formaldehyde resin)

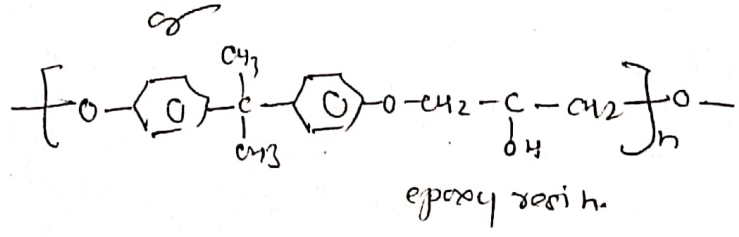
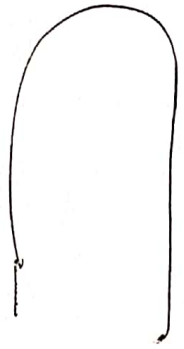
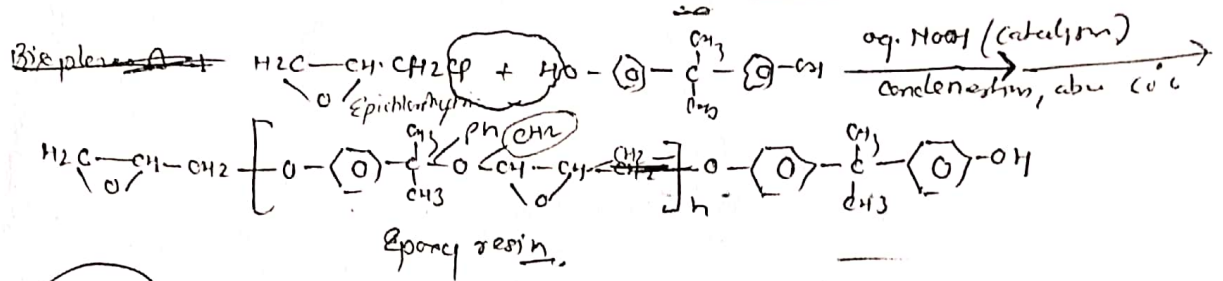
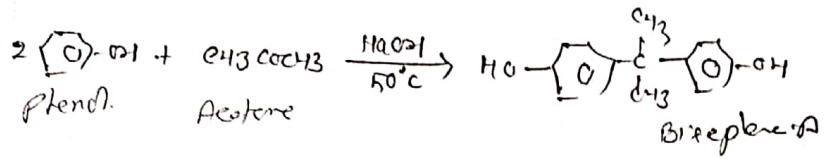


(5) Epoxy Resin: (Polymer)

(a) Synthesis of epichlorohydrin -



(b) Synthesis of bis-phenol-A.



Molecular Weight of Polymers:-

M.W. of a polymer influences various properties

of a polymer such as solution viscosity, softening temp, tensile and impact strength and heat resistance. The low M.W. polymers are generally soft and gum-like resins. The high M.W. polymers are tougher and more heat resistant. Thus, control of molecular weight of a polymer is often important for industrial polymerization. The M.W. of polymer depends upon the number of simple molecules joined together during polymerization reactions i.e. upon the degree of polymerization. But the polymerization chains or loops at different stages, therefore the final product will contain the macromolecules of different weights. Most of the polymers are polydisperse i.e. the molecules of the polymers sample do not have identical molecular weight (M). Hence it is necessary in their case to take average molecular weight. There are two types of average molecular weights:

(i) The no. average molecular weight (M<sub>n</sub>) -

$$M_n = \frac{n_1 M_1 + n_2 M_2 + n_3 M_3 + \dots}{n_1 + n_2 + n_3 + \dots}$$

where n<sub>1</sub>, n<sub>2</sub>, n<sub>3</sub> ... are the numbers of molecules having masses M<sub>1</sub>, M<sub>2</sub>, M<sub>3</sub>, ... , respectively.

$$\sum n_i \bar{m}_i = \frac{\sum n_i m_i}{\sum n_i} \quad \text{--- (1)}$$

(ii) The Weight-Average Molecular Weight (M<sub>w</sub>)

$$\bar{M}_w = \frac{m_1 M_1 + m_2 M_2 + m_3 M_3 + \dots}{m_1 + m_2 + m_3 + \dots}$$

if n<sub>1</sub>, n<sub>2</sub>, ... denote the number of molecules having masses M<sub>1</sub>, M<sub>2</sub>, ... then, m<sub>1</sub> = n<sub>1</sub> M<sub>1</sub>, m<sub>2</sub> = n<sub>2</sub> M<sub>2</sub> - (where ~~n<sub>1</sub>~~ m<sub>1</sub> = n<sub>1</sub> M<sub>1</sub>)

$$\bar{M}_w = \frac{n_1 M_1^2 + n_2 M_2^2 + n_3 M_3^2 + \dots}{n_1 M_1 + n_2 M_2 + n_3 M_3 + \dots}$$

$$\sum \bar{M}_w = \frac{\sum n_i M_i^2}{\sum n_i M_i} = \frac{\sum w_i M_i}{\sum w_i} \quad \text{--- (2)}$$

From (2) it is clear that M<sub>w</sub> is always greater than M<sub>n</sub> for polydisperse systems. For monodisperse systems M<sub>w</sub> = M<sub>n</sub>, thus, the ratio M<sub>w</sub> / M<sub>n</sub> is known as distribution ratio or an index of polydispersity.

(iii) Z-average M.W (M<sub>z</sub>) -

$$M_z = \frac{\sum n_i M_i^3}{\sum n_i M_i^2}$$

--- (3)  $\left. \begin{array}{l} \bar{M}_z > \bar{M}_w > \bar{M}_n \\ \text{i.e. real polymer system} \\ \bar{M}_n : \bar{M}_w : \bar{M}_z = 1 : 1.386 : 1.617 \end{array} \right\}$

~~Q. 1.7~~ ~~→~~ ~~#~~ A polydisperse sample prepared by mixing  
 three monodisperse samples in the following proportions. (2)

- 1g - 10,000 M.W.
- 2g - 50,000 M.W.
- 2g - 1,00,000 M.W.

# A polydisperse sample of polystyrene is prepared by mixing —

Using this above proportion, calculate the number-average (M<sub>n</sub>), weight-average M<sub>w</sub>.

(i)  $M_n = \frac{\sum_{i=1}^3 N_i M_i}{\sum_{i=1}^3 N_i} = \frac{\sum_{i=1}^3 W_i}{\sum_{i=1}^3 \left(\frac{W_i}{M_i}\right)} = \frac{1+2+2}{\left(\frac{1}{10,000} + \frac{2}{50,000} + \frac{2}{1,00,000}\right)} = 31,250$

(ii)  $M_w = \frac{\sum_{i=1}^3 N_i M_i^2}{\sum_{i=1}^3 N_i M_i} = \frac{\sum_{i=1}^3 W_i M_i}{\sum_{i=1}^3 W_i} = \frac{10,000 + 2(50,000) + 2(1,00,000)}{5} = 62,000$

- Q. 1.8
- 1g - 20,000 M.W.
  - 2g - 40,000 M.W.
  - 3g - 80,000 M.W.

$M_n = \frac{1+2+3}{\frac{1}{20,000} + \frac{2}{40,000} + \frac{3}{80,000}} = \frac{6}{0.00005 + 0.00005 + 0.0000375} = \boxed{43636.364}$

$M_w = \frac{(1 \times 20,000) + 2 \times (40,000) + 3 \times (80,000)}{1+2+3} = \boxed{56666.67}$



(i) Molecules exist in next page.

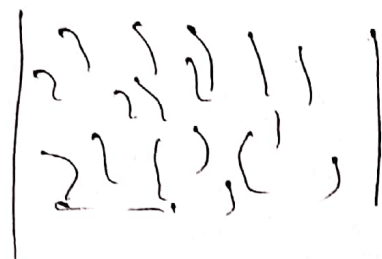
(ii) Secondary Valency forces: - s. Vander Waal forces, affect the mechanical strength of polymers. The strength of these forces increases with the increase polarity and decreases sharply with increasing distance. The individual energies in straight chain and branched chain polymers are low but the cumulative effect of large numbers of these bonds along the polymer chain results in large electrostatic fields of attraction. Such polymers exhibit mechanical strength only when chain length is greater than 150 to 200 atoms in a line. Usually the low m.w.p. are soft, brittle and low heat resistant. The strength of the straight chain polymers depends on the deformation power. Polyvinyl chloride is a hard and stiffer polymer than polyethylene because of polar atom chlorine. The cross-linked polymers are most strong and hard, because of strong covalent-covalent bonds which restrict the slipping of one molecule over the other.

(iii) Mechanical Properties (Deformation)

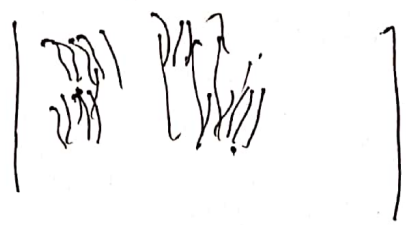
Generally the deformation is found in

thermoelastic polymers on application of heat or pressure or both as they are low heat resistant. At low strain (< 1%), the deformation of most polymers is elasto-elastic meaning that the deformation is homogeneous and full recovery can occur over a finite time. The linear polymers without branched or cross linked structure show the greatest degree of deformation because of the weak intermolecular forces prevail in these polymers. At high temp, the Vander Waal forces become weak and consequently easy deformation occur. While in case of cross-linked and branched polymers which are having more covalent bonds than weaker intermolecular forces, deformation does not occur on heating. These are generally the thermosetting polymers. Such polymers, at higher strains, the deformations are not reversible unless the polymer is heated above its glass transition temp. ( $T_g$ ).

Crystallinity in Polymers :- Some polymers cooled from the melt can organize into regular crystalline structure. The polymers show a ordered structure and called crystalline polymer. The crystalline polymers of high m.w. have perfect organization than crystals of low m.w. crystallized from the solution. The basic units of crystalline polymer morphology are crystalline lamella consisting of arrays of folded chains. Crystalline regions occur when several polymer molecules or strands of polymer molecules are oriented parallel to each other. Under these conditions weak interchain forces hold the molecules together conferring strength on the polymer. On the other hand the polymer molecules are not sufficiently close to each other for the inter-chain forces to operate. These regions are formed as amorphous.



(a) amorphous.



(b) crystalline polymer.



Cryst. polymer embedded in an amorphous matrix.

- Plastic is mainly amorphous and a fibre is a highly crystalline.

Some points for crystalline polymer :-

(1) Some polymers crystallized from the melt or from concentrated solution, crystallites can organize into large spherical structure called spherulites.

(2) Since no polymer is completely crystalline, which may exhibits both a glass transition temp. ( $T_g$ ) corresponding to long range ~~segmented~~ segmented motion in the amorphous regions and a crystalline melting temp ( $T_m$ ) at which crystallites are destroyed and an amorphous disordered melt is formed.

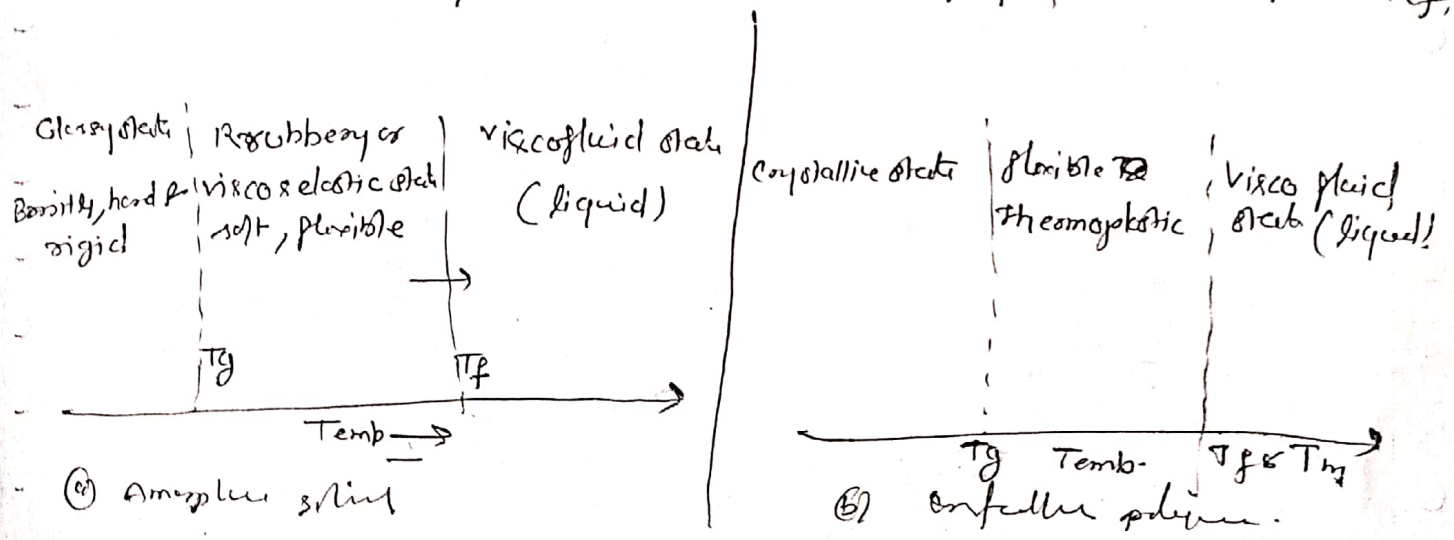
(3) In fibrous polymeric structures, the small aggregated crystallites are not large enough to appear as a small thread like unit are called fibrils.

④ Transparency of crystalline polymer is more or less directly dependent on the size of spherulites. Larger size spherulites decrease the transparency of a given polymer.

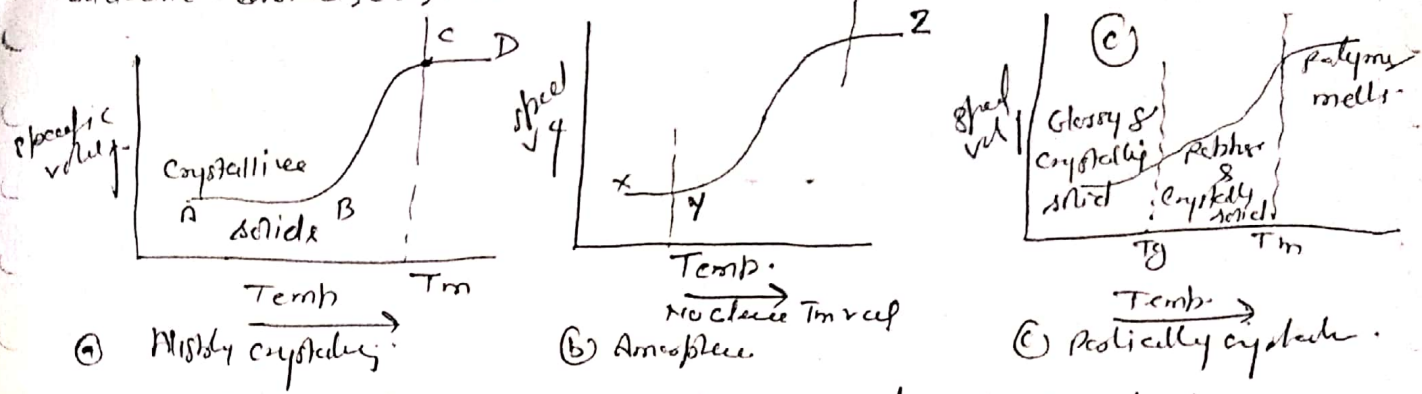
⑤ → The Amorphous state - Completely amorphous polymer exist as long as capable of forming randomly coiled, interpenetrating chain that are free of long range order. The liquid phase state is characterized by the absence of any two long range order. Two situations to have (i) where molecular mobility is frozen, and (ii) where molecular mobility is activated. In both the situations, no long range order is exhibited. When the substance does not possess any molecular mobility, it exists as a physical solid, but it does not possess any long range order, so it is unable to crystallize. i.e. amorphous solids.

An amorphous substance cannot exhibit crystallinity, but can exist as a glassy solid (liquid phase state) or as a molten liquid and the transition from the glassy solid to molten liquid occurs at a temp.

GLASS TRANSITION TEMPERATURE - The temp that marks the transition from the amorphous solid state to the melt state is called the glass transition temp ( $T_g$ ). The  $T_g$  below which polymer is hard, brittle and rigid glassy state and above which it is soft, flexible. The  $T_g$  marks in inorganic state i.e. viscosity increases rapidly as a polymer is cooled from the melt state. At  $T_g$  point, on further heating the polymer without bonded or cross linked ones, becomes a highly viscous liquid and starts flowing. This state is termed as viscofluid state, the transition temp. place at the flow temp.



In case of amorphous polymers, there is no any close melting point. When a polymer changes from one state to another, there is a gradual change in many of its physical (specific volume, specific heat, density, dielectric constant, etc.)...



- The following factors which influence the glass transition temp.
  - (1) Segmental and chain mobilities are easier in non-crystalline polymers than in crystalline polymers.
  - (2) Linear polymers chains made of single bonds have a high degree of freedom for rotation, thus they have higher segmental mobility than the bulkier, aromatic or cyclic structure polymers.
  - (3) The polymers having more secondary valency forces are having higher mobility.
  - (4) Those polymers having polar segments show lower mobility.
    - The  $T_g$  of polymer is influenced by its molecular weight upto around a value of 20,000.
    - $T_m$  is the temp of the crystalline polymer where it melts to the liquid state.

CHEMICAL PROPERTIES OF POLYMERS → The chemical behavior of a polymer determines where it will be crystalline or amorphous.

- (a) Solubility, like dissolves like:
  - (a) polymers of more aliphatic character are more soluble in aliphatic solvents; while aromatic ones are more soluble in aromatic solvents.
- (b) Solubility of polymers in a given solvent decreases with increase in m.w. of polymer as the viscosity of the solution increases.
- (c) Greater the degree of cross-linking in the polymer, lesser is its solubility.
- (d) Polar polymers are soluble in polar solvents like water, alcohol, ketone, etc.
- (e) Non-polar polymers are soluble in non-polar solvents like benzene, toluene, xylene, etc.

M.W. Average - The average molecular weight  $M_i, m_i$  we expect (13)

$$\bar{M} = \frac{\sum_i N_i M_i^\alpha}{\sum_i N_i M_i^{\alpha-1}} \quad \text{--- (1)}$$

When  $N_i$  indicates the number of moles of molecules with a molecular weight of  $M_i$  and the parameter  $\alpha$  is a weighting factor that defines a particular average of the M.W. distribution. The weight  $W_i$  of molecules with molecular weight  $M_i$  is then.

$$W_i = N_i M_i \quad \text{--- (2)}$$

M.W. that are important in determining polymer properties are the number average,  $\bar{M}_n$  ( $\alpha=1$ ), the weight-average,  $\bar{M}_w$  ( $\alpha=2$ ), and the Z-average,  $\bar{M}_z$  ( $\alpha=3$ ), ~~and~~ M.W.



(Fig. ... continuous distribution of M.W. as plot of the no. of moles of chains  $N$ , having M.W.  $M$ , against  $M$ .)

from eqn (1) that the number average M.W. for a discrete distribution of M.W. is given as

$$\bar{M}_n = \frac{\sum_{i=1}^N N_i M_i}{\sum_{i=1}^N N_i} = \frac{\sum_{i=1}^N W_i}{\sum_{i=1}^N (W_i / M_i)}$$

where  $N$  is the total no. of molecular weight species in the distribution. ~~the expression~~ i.e. the number average molecular weight

$$\bar{M}_n = \frac{\int_0^\infty M N dM}{\int_0^\infty N dM}$$

The respective relationships for the weight-average m.w. of a discrete and a continuous distribution ...

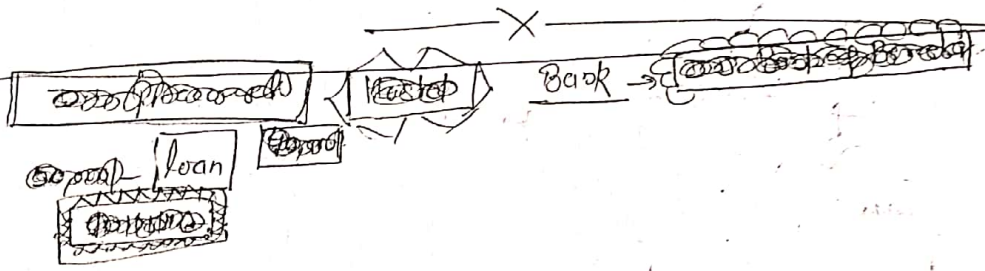
$$\bar{M}_w = \frac{\sum_{i=1}^N n_i M_i^2}{\sum_{i=1}^N n_i M_i} = \frac{\sum_{i=1}^N w_i M_i}{\sum_{i=1}^N w_i}$$

and,

$$\bar{M}_w = \frac{\int_0^{\infty} M^2 dM}{\int_0^{\infty} M dM}$$

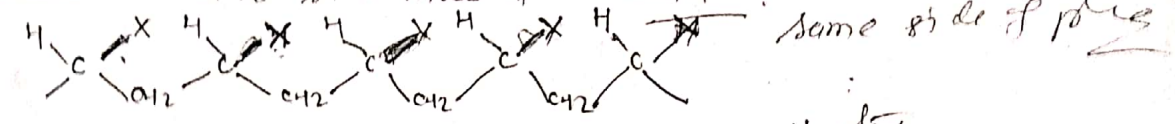
In the case of high-m.w. polymers, the number-average m.w. is determined by membrane osmometry, while the weight-average m.w. is determined by light scattering and other scattering techniques.

To measure of the breadth of the m.w. distribution is given by the ratio of m.w. averages. For this purpose, the most commonly used ratio is  $\bar{M}_w / \bar{M}_n$ , which is called the polydispersity index.

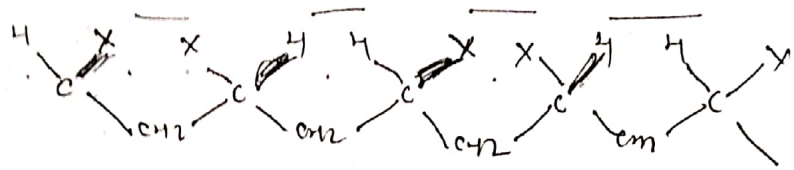


tacticity (stereoregularity) of polymers

(i) Isotactic - The head to tail configuration in which the functional groups are on the same side of the chain

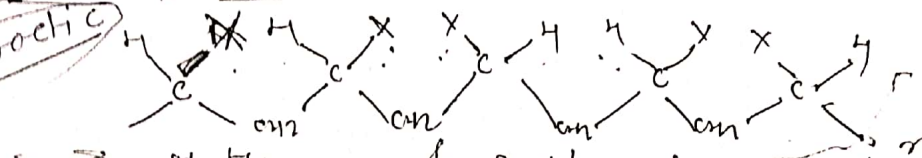


(ii) Syndiotactic, arrangement of side groups are at ~~random~~ <sup>alternately</sup> around the main chain



(iii) Atactic

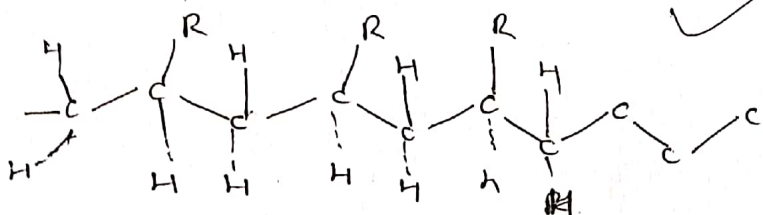
Atactic



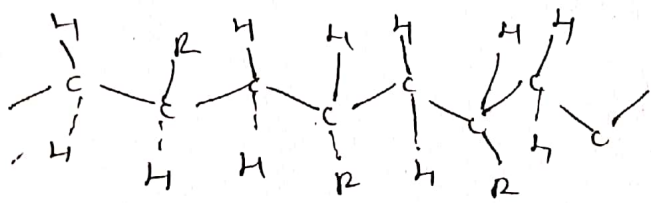
As the arrangement of side groups is in <sup>random</sup> ~~is in~~ alternating fashion, it is called syndiotactic

Tacticity → the orientation of monomer units in a polymer molecule can take place in an orderly (tacticity) or disorderly fashion with respect to main chain. The difference in configuration affects their physical properties.

(i) Isotactic → The arrangement of functional groups are all on the same side of the chain.



(ii) Atactic → arrangement of functional groups are at random around the main chain, it is called atactic.



(iii) Syndiotactic → The arrangement of side groups is in alternating fashion,

